Study of ZnO, SnO₂ and Compounds ZTO Structures Synthesized for Gas-Detection

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Abstract

Semiconductor-based metal oxide gas detector of five mixed Z:S from zinc chloride salt (0,25,50,75,100%) ratios with tin chloride salt, were fabricated on glass substrate by a spray pyrolysis technique with thickness were about ($0.2\pm0.05~\mu m$) using water soluble as precursors at a substrate temperature 500 C°±5, 0.05 M ,and their gas sensing properties toward (CO $_2$, NO $_2$ and SO $_2$ gas at different concentration (10,100,1000 ppm) in air were investigated at room temperature which related with the petroleum industry.

Furthermore structural and morphology properties was inspecting. Experimental results show that the mixing ratio affect the composition of formative oxides (ZnO,Zn₂SnO₄,Zn₂SnO₄+ZnSnO₃,ZnSnO₃, SnO₂) ratios mentioned in the above respectively, and related with the sensitivity of the tested oxidation gases.

Key Word: Zinc stannate, gas sensor, ternary metal oxides, spray pyrolyess, pollutant gases, XRD.

دراسة تراكيب SnO2,ZnO و مركباتها ZTO المؤلفة على الكشف الغازي

الخلاصة.

في هذا البحث حضر كاشف شبه موصل اوكسيدي من خمس خلطات Z:S من ملح كلوريد الزنك (100, 0.2) (500, 25,0%, 750, 25,0%) (500, 25,0%, 750, 25,0%) و بسمك حوالي (\pm 0.05 µm (\pm 0.05) على قواعد من الزجاج و باستخدام الماء كمذيب , و بدرجات حرارة ترسيب بلغت (\pm 0.05) و بتركيز \pm 0.05 (\pm 0.05) ككاشف للغازات الملوثة مثل (\pm 0.02, CO و بتركيز (\pm 0.05) و بتركيز (\pm 0.05) و بدرجة حرارة الغرفة والمرتبطة مع الصناعة النفطية .

تم تشخيص التبلور و مورفولوجية سطح الأغشية المحضرة بواسطة قياسات حيود الأشعة السينية XRD و المجهر الالكتروني الماسح SEM ومجهرة القوة الذري AFM لها . بينت النتائج أن نسبة الخلط تؤثر على تبلور الأكاسيد المتكونة و المركبات الثلاثية المتكونة حيث كانت(ZnO,Zn₂SnO₄,Zn₂SnO₄+ZnSnO₃,ZnSnO₃, SnO₂) للنسب المذكورة في أعلاه على التوالي وترتبط مع حساسية الغازات المؤكسدة المفحوصة.

الكلمات المرشدة: قصديرات الخار صين, كاشف غازي, اكاسيد معادن ثلاثية, غازات ملوثة, حيود الاشعة السينية.

INTRODUCTION

he development of high performance solid state chemical sensors is nowadays of outmost importance in many advanced technological fields[1]. Numerous researches have shown that a characteristic of solid-state gas sensors is the reversible interaction of the gas with the surface of a solid-state material. In addition to the conductivity change of gas-sensing material, the detection of this reaction can be performed by measuring the change of capacitance, work function, mass, optical characteristics or reaction energy released by the gas/solid interaction. Various materials, synthesized in the form of porous ceramic, and deposited in the form of thick or thin films, are used as active layers in such gassensing devices [2]. Zinc stannate or zinc tin oxide (ZTO) is a class of ternary oxides that are known for their stable properties under extreme conditions, higher electron mobility compared to its binary counterparts and other interesting properties. These materials is thus ideal for applications from, gas detector, solar cells photocatalysts, light-emitting diodes, field effect transistors and (heterojunction and homojunction) diodes. In 2000, K. Zakrzewska, [3] study mixed-oxide systems such as SnO₂-WO₃, TiO₂-WO₃ and SnO₂ -TiO₂ with the ultimate aim of the application in gas sensing devices and study dynamic changes in the electrical resistance (R) of SnO₂-WO₃ thin films for samples deposited at two different substrate temperatures, 100 and 300C° upon exposure it to hydrogen at 400°C and CH₄ at (350, 700°C). In 2006, S. M. Chou, L. G. Teoh, et. al., [4] study ZnO:Al thin film gas sensor for detection of ethanol vapor (400 ppm) at 250 C°. In 2007, M.Y. Faizah, A. Fakhru'l-Razi, et.al., [5] study gas sensor application of carbon nano tubes for (H₂, CO₂ and NH₃) gases at temperature higher than 200 C°.

In 2008, L. Al-Mashat, H. D. Tran, et. al., [6] study conductometric hydrogen gas sensor based on polypyrrole (PPy) nanofibers at temperature below (100 $^{\circ}$). The goal of this work is to fabricating different semiconductor-based metal oxide gas detector of five mixed film by spray pyrolysis technique, and study some of their structural properties and sensitivity to some pollutant gases.

Experimental Work

The work includes the following steps;

The First Step:

- Preparation of an aqueous solution of zinc chloride ($ZnCl_2$ and $SnCl_2.2H_2O$) with purities 99.9%, at (0,25,50,75,100)% ratios .
- The concentration was (0.05 M), the acidity was maintained to be \approx 5.5 pH during spraying .The spraying apparatus was manufactured locally in the university laboratories. In this technique, the prepared aqueous solutions were atomized by a special nozzle glass sprayer at heated collector glass fixed at thermostatic controlled hot plate heater.
- Air was used as a carrier gas to atomize the spray solution with the help of an air compressor with pressure (7 Bar) air flow rate (8 cm³/sec) at room temperature.
- The glass substrate(0.1 cm thickness) temperature was maintained at 500 °C during spraying, coating thickness was(0.2) µm by tested by optical microscopy.
- Atomization rate was (1 nm/s) with (2.5 ml/min) of feeding rate. The distance between the collector and spray nozzle was kept at (30 ± 1 cm) the volume of spray solution was 50 ml, number of spraying (100), time between two spraying (10 sec).

• The spray of the aqueous solution yields the following chemical reaction [7,8,9]:

The Second Step:

Spraying the solution on the substrate at temperature $(500 \pm 5^{\circ}C)[10,11]$.

The Third Step: Annealing samples, by using furnace(type Nabertherm) at temperature 500°C for (60 min) and cooling inside furnace.

The Forth Step:

Inspection which include:

- 1- X-ray diffraction by diffractometer type with radiation $CuK\alpha$ ($\lambda = 1.5406 \text{ Å}$). This inspection was carried out in Advanced Materials Research Center at the Technology and Science Ministry, with continuous scanning.
- 2- Scanning Electron Microscopy (SEM):The SEM study has been carried out by Electron Gun Tungsten heated filament, Resolution 3 nm at 30kV, Accelerating voltage 200 V to 30kV, chamber internal size: 160 mm (Japan), with Au coating for (20 sec). The setting in Nano Center at University of Technology.
- 3- Atomic Force Microscopy (AFM): It was taken with a digital instruments, typical data has been taken from AFM height images include root mean square (RMS) and roughness. Made in USA, model AA3000 220V.
- 4- gas-sensing: The gas-sensing experiments were carried out by introducing the thus prepared devices into a home-made test cell, which was consist of a cylinder with cover to restrict prepared gas as in figure (1).

The gas was obtained from reaction solution to rising predicted gases. pollutant gases that prepared CO_2 , NO_2 and SO_2 gases as will be explained in the following produce from reactions equation: [7,10]

The gases -sensing properties were determined at different temperature by measurement of the electrical conductivity of the samples which exposed to various concentration of the gases (10, 100,1000) ppm $_v$ [12,13]. The sensor response was defined using the following equation[14,15]:

Sensor response (%) =
$$[(Ra-Rg)/Ra] \times 100\%$$
 (8)
Where

Ra and Rg are the electric resistance in air and test gas, respectively.

Results And Discussion

1- X-ray results, the result of X-ray diffraction which represent all mixing ratio for precipitation thin films on glass substrates are shows in Fig.(2) the XRD patterns result shows match with standard value in (JCPDS card No. 36-1451). ZnO have wurtzite type polycrystalline thin film with a hexagonal system revealing that the highly dominant to c-axis oriented the film was grown, no characteristic peaks for any other impurities were observed, suggesting sample have high purity. The *c*-axis orientation in ZnO films can be understood by the "survival of the fastest" model, in where nuclei having the fastest growth rate can survive, i.e., *c*-axis orientation is achieved [16,17].

In Fig.(3) it can be seen that the mixing (ZTO) of components di-zinc stannate Zn₂SnO₄ and scant traces of zinc stannate ZnSnO₃, according to standard cards (JCPDS card No.24-1470 and No.28-1486) respectively. The Zn₂SnO₄ have high appearance comparing with ZnSnO₃ because of large accourtement of Zn ion in mixing [18]. All of the diffraction peaks well match the standard diffraction pattern of cubic spinal structure for Zn₂SnO₄ , while ZnSnO₃ has a perovskite face-centered structure, coinciding with [19,20,21]. A different mixing ratio of (ZTO) is shown in Fig. (4) of components zinc stannate (ZnSnO₃) and di-zinic stannate (Zn₂SnO₄) return to show, by more than one peak the mixture homogenized showing deposition efficiency as well as the proper role of annealing in the formation of both two compounds according to stander cards. Another mixing ratio (Z_{0.25}:S_{0.75}) is shown in fig.(5), ZnSnO₃ appears and it has high appearance comparing with Zn₂SnO₄ because of large amount of supplying of Sn ions in mixing. The highest peaks for the compound $ZnSnO_3$, and for Zn_2SnO_4 more than one weak peaks was observed as shown in table (1). Finally in Fig. (6), which is represent mixing ratio of salts (0% zinc oxide and 100% of the tin oxide), SnO₂ pattern of film was much with standard card (JCPDS No.41-1445) have tetragonal rutile structure, no characteristic peaks for any other oxides or impurities were observed, suggesting sample have high purity oxide. Table (1) shows results Data of XRD $Z_1:S_0$ and $Z_0:S_1$ and table (2) Results XRD Data for other Z:S mixing ratios.

2- SEM: Micrographe

Fig. (7) shows ZnO (2D-images), the film deposited with zinc is formed a hexagonal, this is consistent with XRD analysis. The latter corresponds to the diffraction plane (002) in the hexagonal wurtzite structure of ZnO. The appearance of the preferential diffraction peak, assigned to the plane (002), indicates that the film growth is achieved along the axis c of the hexagonal structure normal to the substrate surface as show in Fig (7), this is good harmony with the SEM observation for (N. Lehraki et al) [21]. The ortho stable form of zinc stannate (Zn₂SnO₄) have tri-symmetric facecenter cubic spinel structure, SEM (2-D) images for growth Zn₂SnO₄ show the morphology of microcubes at different magnifications Fig.(8), this is good concordance with the SEM observation for (S. Baruah and J. Dutta) [8]. SEM (2-D) images for growth Zn₂SnO₄ and ZnSnO₃ microcubes at different magnifications are shown in Fig.(9). Zinc stannate (ZnSnO₃) has structure perovskite, according to results XRD which shows almost all the particles are spherical in shape leaving some space between them as shown in Fig. (10), this is good matching with the SEM observation for (R. Singh et. al.) [22]. SEM (2D-images) obtained for $Z_0:S_1$

represented by SnO_2 pure film for sensor application are shown in Fig.(11), summarizes the scanning electron microscopy (SEM) analysis of the SnO_2 deposit have a large area from the surface of the deposit is homogeneous [23].

3- AFM result

AFM is powerful technique to investigate the surface morphology at nano to microscale. The surfaces of the films are shown in Figs.(12,13,14,15,16) for mixing (1, 0.75, 0.5, 0.25, 0)% of Z respectively. Irregularities in the film surface (top view) from the spray technique, which benefited tremendously for the gas sensing properties [24]. The surfaces morphology of the ZnO films as observed from the (3-D) AFM micrograph figs. (12,13,14,15,16) confirms that the grains were uniformly distributed within the scanning area (2014 X 2014) nm. This surface characteristic is important for applications gas sensors [26]. The roughness was found 19.62 nm and average diameter 76.06 nm, For Zn₂SnO₄. In fig.(13) the roughness was 18.14 nm and average diameter 91.33 nm, and in fig.(14) the roughness was found 6.72 nm and average diameter 109.99 nm, fig.15 shows ZnSnO₃ surface with roughness was found 4.80 nm and average diameter 81.20 nm .Fig.16 shows surface morphology of SnO₂ with roughness 22.94 nm and average diameter 68.31 nm.

4- Gas Sensor Results

For the investigation of the oxidation gases sensing properties of the Z:S oxide films, the optimum operation gas temperature should be fixed at the first. Fig. (17), shows an increasing the response of sensor by increased SO_2 gas concentration on the one hand and increased in response with variation of Z:S oxide ratio ,which resulted in change oxides consisting as it was seen in XRD results, the difference broad here since low concentrations of the gas between the pure oxides and mixtures are obey to the same relationship for oxidation gases[26].

$$O_2(g) + e^- \rightarrow O_2^- (ads)$$
(9)

It can be observed that sensors based on pure oxide ZnO and SnO $_2$ show relatively low response at the same operating temperature in the range from (10 - 1000) ppm, with the maximum response of (88. 39 %) and(90.87 %) for ZnO and SnO $_2$ at 1000 ppm, respectively. In contrast, sensor with mixed oxides (Zn_2SnO_4 , $ZnSnO_3$) exhibits an increase of response to reach the maximum value of 100% at the 1000 pm concentration of SO_2 gas for them . Fig.(18) shows the response time as a function of gas concentration ,it can be seen that the time diminish by increased gas concentration[27-28].

In the presence of NO_2 in air the sample's sensitivity is shown in Fig.(19) acceptor gas adsorption[29]:

$$NO_{2(gas)} + \overline{e} \leftrightarrow N\overline{O}_{2(ads)}$$
 ...(10)

The same equation (9)

where:

NO₂ :molecule in the gas phase

e = : an electron from the conduction band of the semiconductor

NO⁻_{2(ads)}: adsorbed form of NO₂ on the semiconductor surface.

This indicates the role of the electronic factor in the sensor sensitivity. After the adsorption on the surface the NO₂ molecule acts as an electronic trap, profoundly depleting the electron density in the conduction band and leading to a significant

decrease of the crystalline material conductance , and this indicates the role of the electronic factor in the sensor sensitivity[30]. So, an increase of concentration of electrons, which have enough energy to overcome the electric barrier resulting from the negative charging of the surface, favors reaction that leads to an increase of the sensor signal. It can be observed that sensors based on pure oxide ZnO and SnO $_2$ show relatively low response at the same operating temperature in the range from (10 - 1000)ppm , with the maximum response of (80. 0 %) and (84.44 %) for ZnO and SnO $_2$ at 1000 ppm , respectively. In contrast, sensor with mixed oxides (Zn $_2$ SnO $_4$, ZnSnO $_3$) exhibits an increase of response to reach the maximum value of (90.97%) at the 1000 pm concentration of NO $_2$ gas for them .

Fig.(20) shows the response time as a function of gas concentration, it can be seen that the time diminish by increased gas concentration. And it is higher for pure oxide and fewer for mixed oxides.

Fig.(21) shows an increasing the response of sensor by increased CO₂ gas concentration on the one hand and an increased in response with variation of Z:S Oxide ratio. As shown in the figure, when O₂ molecules are adsorbed on the surface of metal oxides, they would extract electrons from the conduction band Ec and trap the electrons at the surface in the form of ions, differences in the nature of the interaction with different gas CO2 from NO2 and SO2gas on the one hand the length of the bond, and enthalpy .Also the influence of the crystallite which found in sensitivity[29,30]. This will lead a band bending and an electron depleted region. The electron-depleted region is so called space-charge layer, of which thickness is the length of band bending region. Reaction of these oxygen species with reducing gases or a competitive adsorption and replacement of the adsorbed oxygen by other molecules decreases and can reverse the band bending, resulting in an increased conductivity. O- is believed to be dominant at the operating temperature which is the work temperature for most metal oxide gas sensors [21,31] .It can be observed that sensors based on pure oxide ZnO and SnO₂ show relatively low response at the same operating temperature in the range from (10 - 1000) ppm, with the maximum response of (78. 48 %) and (80.54 %) for ZnO and SnO₂ at 1000 ppm, respectively. In contrast, sensor with mixed oxides (Zn₂SnO₄, ZnSnO₃) exhibits an increase of response to reach the maximum value of (91.62%) at the 1000 pm concentration of CO₂ gas for them . Fig.(22) shows the response time as a function of gas concentration ,it can be seen that the time diminish by increased gas concentration. And it is higher for pure oxide and fewer for mixed oxides. Less response time was (40-17 sec) for (10-1000 ppm), starting response times for the pure oxide films(ZnO,SnO₂) were about (9 sec) and for mixed oxide less than (4 sec).

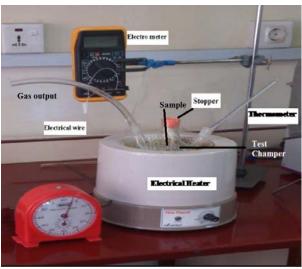
Table (1) Results data of XRD $Z_1:S_0$ and $Z_0:S_1$

Mix	Comp.	$2\theta_{ST}$	$2\theta_{\rm M}$	$I_{ST}\%$	$I_M\%$	hkl
$Z_1:S_0$	ZnO	31.770	31.762	57	13	100
		34.422	34.409	44	100	002
		36.253	36.248	100	9	101
Z_0 : S_1	SnO_2	26.611	26.5586	100	100	110
		33.893	33.8393	75	54	101
		37.950	37.8356	21	17	200
		51.781	51.6776	57	38	211

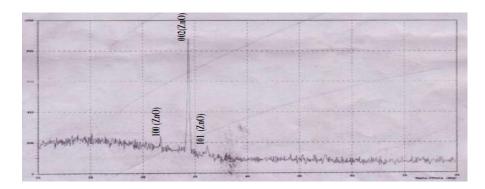
54.759	54.6263	14	11	220

Table (2) Results XRD data of other Z:S mixing ratio.

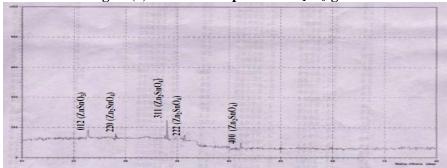
Tuble (2) Results Title data of other 2.5 maning rates									
Comp.	$2\theta_{ST}$	I _{ST} %	hkl	$Z_{0.75}:S_{0.25}$		$Z_{0.5}:S_{0.5}$		$Z_{0.25}$: $S_{0.75}$	
				$2\theta_{M}$	I _M %	$2\theta_{\mathrm{M}}$	I _M %	$2\theta_{\mathrm{M}}$	I _M %
$\mathrm{Zn_2SnO_4}$	29.141	19	220	29.1028	20	29.1028	20	29.1445	20
	34.291	100	311	34.1384	100	34.1384	100	34.2381	20
	35.907	20	222	35.6376	20	35.6376	20	35.2377	17
	41.685	25	400	41.2080	22	41.2080	22	41.5911	17
	55.116	30	511	1		55.1061	30		
ZnSnO3	26.507	100	012	26.3902	22	26.4140	39	26.4357	100
	33.797	50	110			33.5053	61	33.6520	44
	37.768	20	015			37.7356	39	37.6106	31
	51.596	45	116			51.5212	30		
	57.559	3	018			57.5252	13		



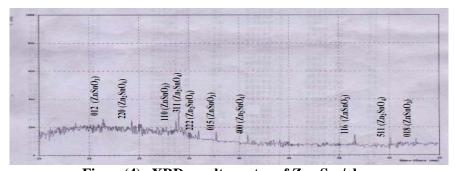
Figure(1) Gas detection measurement system



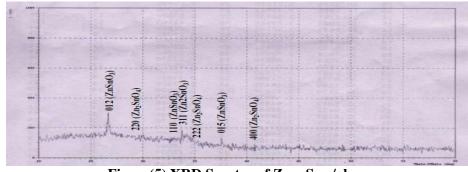
Figure(2) XRD result spectra of $Z_1:S_0/glass$



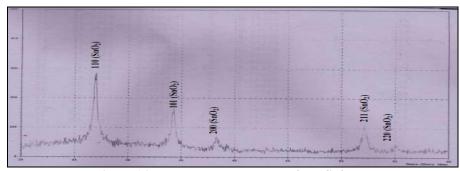
Figure(3) XRD result spectra of Z_{0.75}:S_{0.25}/glass



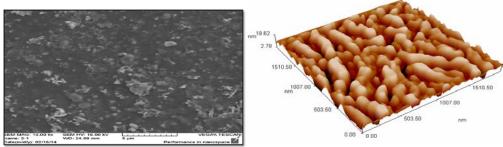
Figure(4) XRD result spectra of $Z_{0.5}$:S_{0.5}/glass



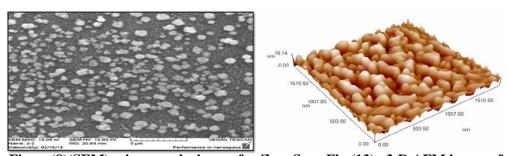
Figure(5) XRD Spectra of $Z_{0.25}$: $S_{0.75}$ /glass



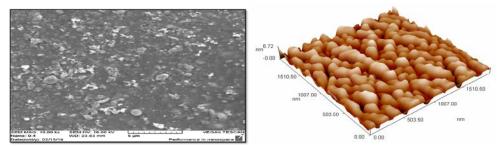
Figure(6) XRD result spectra of Z₀:S₁/glass



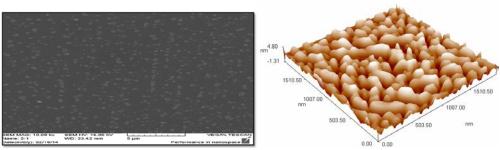
Figure(7) (SEM) micrographs image for Z_1 : S_0 Fig.(12) 3-D AFM image of the Z_1 : S_0



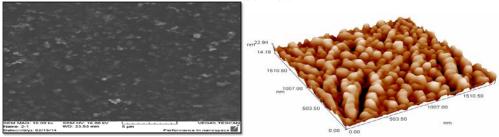
Figure(8)(SEM) micrographs image for $Z_{0.75}$: $S_{0.25}$ Fig.(13) 3-D AFM image of the $Z_{0.75}$: $S_{0.25}$



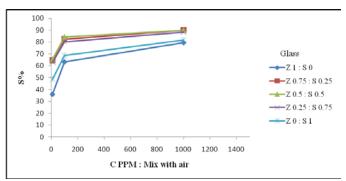
Figure(9)SEM micrographs image for $Z_{0.5}$: $S_{0.5}$ Fig.(14) 3-D AFM image of the $Z_{0.5}$: $S_{0.5}$



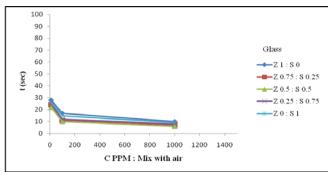
Figure(10) SEM micrographs image for $Z_{0.25}$: $S_{0.75}$ Fig.(15) 3-D AFM image of the $Z_{0.25}$: $S_{0.75}$



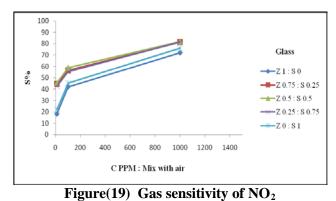
Figure(11)SEM micrographs image for Z_0 : S_1 Fig.(16) 3-D AFM image of the Z_0 : S_1



Figure(17) Gas sensitivity for SO₂



Figure(18) Response time for SO₂



90 80 70 Glass 60 50 →Z1:S0 40 Z 0.75 : S 0.25 30 <u></u>

± Z 0.5 : S 0.5 20 → Z 0.25 : S 0.75 10 Z0:S1 0 1000 1200 1400 C PPM : Mix with air

Figure (20) Response time for NO_2

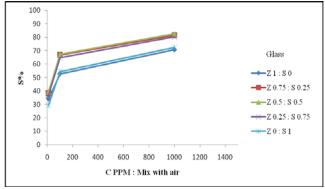
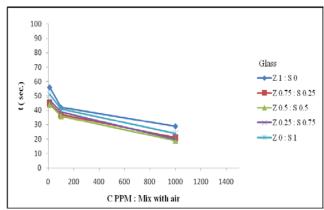


Figure (21) Gas sensitivity for CO₂



Figure(22) Response time for CO₂ gas

Conclusions

- 1- Preparing gas sensor with several oxides form of (ZnO, Zn₂SnO₄, ZnSnO₃, SnO₂) on glass substrates to detect different gases.
- 2- Mixing ratios chloride salts generate oxide compounds (triple and binary) with different responses for different testing gases.
- 3- Homogenous surface of thin films and crystalline structures; hexagonal, tetragonal and cubic for oxides ZnO, SnO_2 , $ZnSnO_3$ and Zn_2SnO_4 respectively. Mixing oxides (Zn_2SnO_4 + $ZnSnO_3$) showed the highest response rate for all testing gases at the best possible time.

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