

Enhancement of Porous Silicon Formation by Using laser Irradiation

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ABSTRACT

In this work porous silicon where prepared by chemical etching assisted with laser. The structural and optical properties of porous silicon are investigated using atomic force microscopy (AFM) and FTIR spectroscopy. FTIR spectrum exhibit the formation of SiH_x (x=1,2) and Si-O bonds. The atomic force microscopy AFM investigation shows the surface roughness (RMS observed was 1.52nm with laser and 1.86 nm without laser) and pyramid like hillocks surface on entire surface which can be regarded as a condensation point to form small skeleton clusters which plays an important role for the strong visible luminescence.

Keywords: porous silicon, chemical etching, optical properties, structure Properties.

تحسين تشكيل السيليكون المسامي باستخدام أشعة الليزر

الخلاصة

في هذا العمل ، حضر سيليكون مسامي بطريقه كيميائيه وبمساعده الليزر . تم دراسه الخصائص البصريه والتركيبه للسيليكون المسامي باستخدام مجهر القوى الذرية ومطياف FTIR. بين طيف FTIR تشكيل SiH_x (x = 1,2) والاصره O-S. يظهر مجهر القوة الذرية AFM خشونة السطح (لوحظ ان RMS 1,52 nm مع ليزر و 1.86 nm بدون الليزر). وهرم مثل سطح التلال على كامل السطح والتي يمكن اعتبارها نقطة تكاثف لتشكيل مجموعات الصغيرة والتي تلعب دورا قويا في الاضاء المرئيه.

INTRODUCTION

Silicon has been attracted great attention with experimental and theoretical interest in last year's, not only as an interesting material for variety of useful optical and electronic devices [1]. The highly perfect crystalline structure of extremely pure silicon single crystals is considered as the starting point for many metro-logical applications [2]. Research interest in all silicon based light – emitting devices (LED) is stimulated by observation of efficient light emission in structures

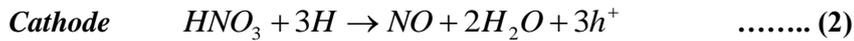
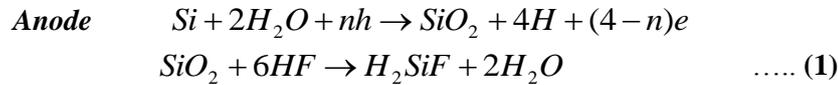
of nanocrystalline silicon [3]. Porous silicon can be considered as silicon crystals have a network of voids in it [4]. The nanosized voids in the silicon bulk result in a sponge-like structure of porous and channels surrounded with a skeleton of crystalline silicon nanowires [5]. The physical properties of porous silicon are fundamentally determined by the shape and diameter of pores, thickness, and relative content of silicon, voids and in some cases, the relative content of different silicon compounds in the formed porous layer [6]. Porous silicon is usually prepared by electrochemical etching (p-Si) under an anodic bias or photochemical etching (n-Si) under light illumination [7]. In addition to electrochemical etching the method of stain etching (SE) [8-10] for formation of porous silicon layer has been developed. Along with its simplicity the stain etching method is interesting because it does not require special equipment and gives the possibility to prepare very thin (≤ 100 nm) porous silicon layer [11]. Fathauer *et al.* carried out stain etching of silicon in HF:HNO₃:H₂O solutions and obtained porous layers similar to those produced by anodic etching [12]. The observation of visible room temperature photoluminescence (PL) from porous silicon for understanding the optical properties of silicon. Since the discovery of visible photoluminescence (PL) at room temperature in porous silicon in 1990 many nanostructured silicon systems have been fabricated and investigated aiming to obtain applicable silicon-based light emitting devices. Although, it yields high-efficiency red-light emission. Porous silicon has considerable attention for solar cell and sensor applications [13]. Its luminescence properties, large surface area, and compatibility with silicon-based devices are good reasons for used in the sensors and the solar cells. Related to these applications, the ability of adjustment and control of some parameters are more important. The energy band gap and the thermal diffusivity are two important parameters in sensors, solar cells [14, 15], electronic [16] and optoelectronic [17, 18] devices, thermal flow sensors [19], isolators [20] and fuel cells [21, 22]. In this work, study the properties of porous silicon prepared by stain etching with and without laser by using FTIR and AFM microscopy.

EXPERIMENTAL WORK

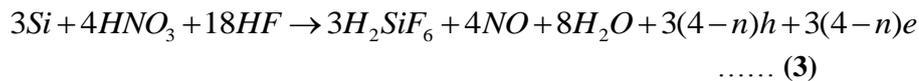
The silicon wafer that was used is n-type (111) with resistivity about (1.5- 4 Ω cm) and (500 ± 15 μ m) thickness. It is rinsed in acetone and ethanol in order to remove dirt and oil, while native oxide layer removed by etching in dilute (1:10) HF: H₂O.

Porous silicon was prepared by chemical etching using an electrolyte containing 40% HF and HNO₃ (1: 3) acids for different etching time (2, 4, 6, 8, 10 & 12) min with and without laser (10 mW with $\lambda=532$ nm), then samples are rinsed in ethanol and dried with a jet of nitrogen gas and stored in a container filled with a methanol to avoid the formation of oxides layer above the porous layer.

chemical stain – etch consisting essentially of HF and nitric acids. The anodic reaction consists mainly of the dissolution of Si, while the cathode reaction is a complicated reduction of HNO₃ which causes holes to be injected into the Si. The proposed anode and cathode reactions as well as the overall reaction are listed below: [23]



Overall



Where *n* is the average number of holes required to dissociate one Si atom. The anode and cathode sites are not necessary fixed during the etching process. HNO₃ is more likely to attack imperfect sites on the crystalline Si surface, such as dislocation, grain boundaries, etc. According to the reactions above, the area where HNO₃ attacks will become local cathode and causes hole injection into the Si. If the injected holes react with Si, the Si atoms will be dissolved and form SiO₂ as shown in equation (1) and thereby turn the original local cathode site into an anode site. The SiO₂ subsequently reacts with HF to form water soluble H₂SiF₆. The residual H⁺ at anode site might react with HNO₃ and the cycle repeats.

The surface morphology of stained sample was examined by using optical microscopy (Olympic), and atomic force microscope AFM (Digital instruments nanoscope). the chemical composition of surface was best probed with Fourier Transform Infrared (FTIR) Spectroscopy (8400S, SHIMADZU), which Scans over range between (400-4000) cm⁻¹.

Porosity and thickness of the samples were calculated by the gravimetric method [17]. The virgin wafer is first weighed before anodization (m₁) then just after anodisation (m₂) and finally after dissolution of the whole porous layer in a molar NaOH aqueous solution (m₃). Uniform and rapid stripping in the NaOH solution is obtained when the PS layer is covered with a small amount of ethanol which improves the infiltration of the aqueous NaOH in the pores. The porosity is given simply by the following equation (18):

$$P(\%) = \frac{m_1 - m_2}{m_1 - m_3} \times 100 \quad \dots (4)$$

$$d = \frac{m_1 - m_3}{\rho S} \quad \dots (5)$$

Where ρ is the Si density and S is surface area.

RESULTS AND DISCUSSION

The FTIR spectra of p-type porous silicon shown in Figure (1) for etching time (2 & 12) min. The band between “1000 to 1250 cm⁻¹” is the Si-O-Si stretching

vibration. So that Si-O-Si structure with vacancies ($\nu(\text{Si-O-Si}) = 1080 \text{ cm}^{-1}$) called “Not bridge oxygen hole center” (NBOHC), as the surface structure responsible for PL emission. A peak at $\sim 979 \text{ cm}^{-1}$ is related to SiH bending vibration, while a peak at $\sim 918 \text{ cm}^{-1}$ originates from a bond between Si and F and it is a Si-F₂ symmetric stretching mode. Then a peak $\sim 902 \text{ cm}^{-1}$ is SiH bending vibration. A peak at $\sim 856 \text{ cm}^{-1}$ is related to SiH₂ wagging vibration mode, while a peak at $\sim 840 \text{ cm}^{-1}$ is related to Si-O-Si and a peak at $\sim 862 \text{ cm}^{-1}$ is due to SiH₃ symmetric bond, then a peak at $\sim 664 \text{ cm}^{-1}$ is due to SiH deformation mode and peak at $\sim 611 \text{ cm}^{-1}$ is related to the Si-Si bond vibration. The peak around $\sim 750 \text{ cm}^{-1}$, $\sim 1090 \text{ cm}^{-1}$ and $\sim 1438 \text{ cm}^{-1}$ is related to NO₃ wagging vibration modes, while peaks at $\sim 1460 \text{ cm}^{-1}$, $\sim 846 \text{ cm}^{-1}$ and $\sim 831 \text{ cm}^{-1}$ are corresponding to hydrocarbon vibration mode. While peaks at $\sim 465 \text{ cm}^{-1}$ and $\sim 470 \text{ cm}^{-1}$ are related to SiO-Si mode. A peak at $\sim 856 \text{ cm}^{-1}$ is related to SiH₂ wagging vibration mode.

AFM image

Figure (2) shows AFM image for PS prepared at 2 min with (fig.2a) and without laser Figure (2b). These AFM images reveal a roughened surface with roundish microstructures. The root-mean-squares (rms) observed are 11.7 nm (fig.2a) and 6.38 nm Figure (2b). The lateral sizes of the microstructures are also found to be in the range of 50-280 nm Figure (2a) and 80-420 nm Figure (2b).

Porosity and Thickness

Figure (3) shows the variation of thickness Figure (3a) and porosity Figure (3b) with etching time for prepared sample with and without laser. It can be noted that thickness of PSi layers was grows linearly with increasing the etching time, also the porous layer for sample prepared with laser was higher than sample prepared without laser by ~ 4 times. While, The porosity initially increases rapidly with increasing etching time, also the porosity for sample prepared with laser was higher than sample prepared without laser by $\sim 1.675\%$.

Etching rate

Figure (4) show the variation of etching rate with etching time for prepared sample with and without laser. The etching rate was determined by measuring the maximum PSi layer thickness at the spot center. The etching rate could be calculated by the relation [23].

Maximum etching rate = maximum thickness of PSi (μm) / irradiation time (min)

Which found that the etching rate increasing for different etching time as shown in Figure (4). this result related to thickness of porous layer.

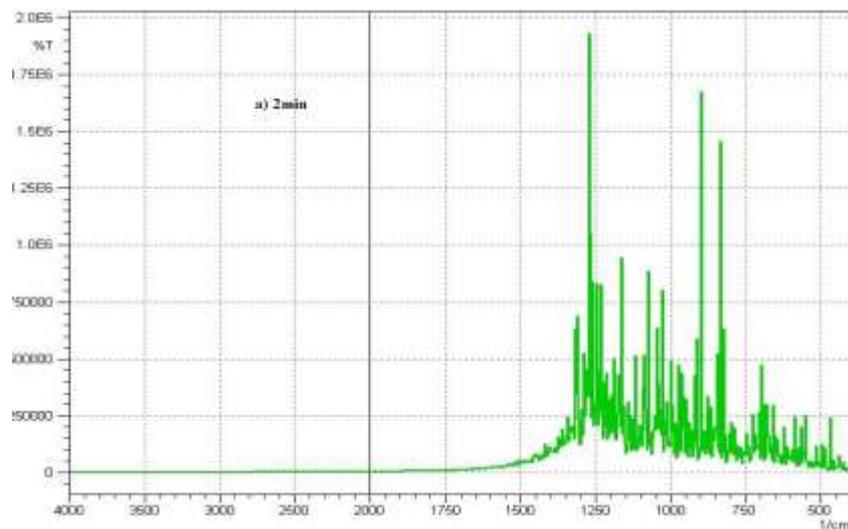
CONCLUSIONS

Porous silicon was prepared by stain etching in HF/HNO₃ with and without laser.. It is obvious that the etching process is being efficient when the surface have be irradiation with laser. Increasing the etching time leads to create surface has fine shapes with large thickness. These shapes have significant effects on the etching rate as well as the porous silicon formation.

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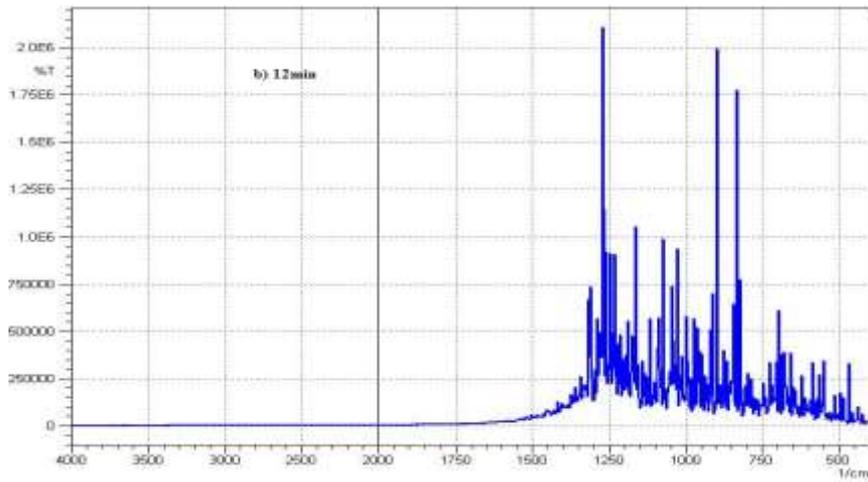
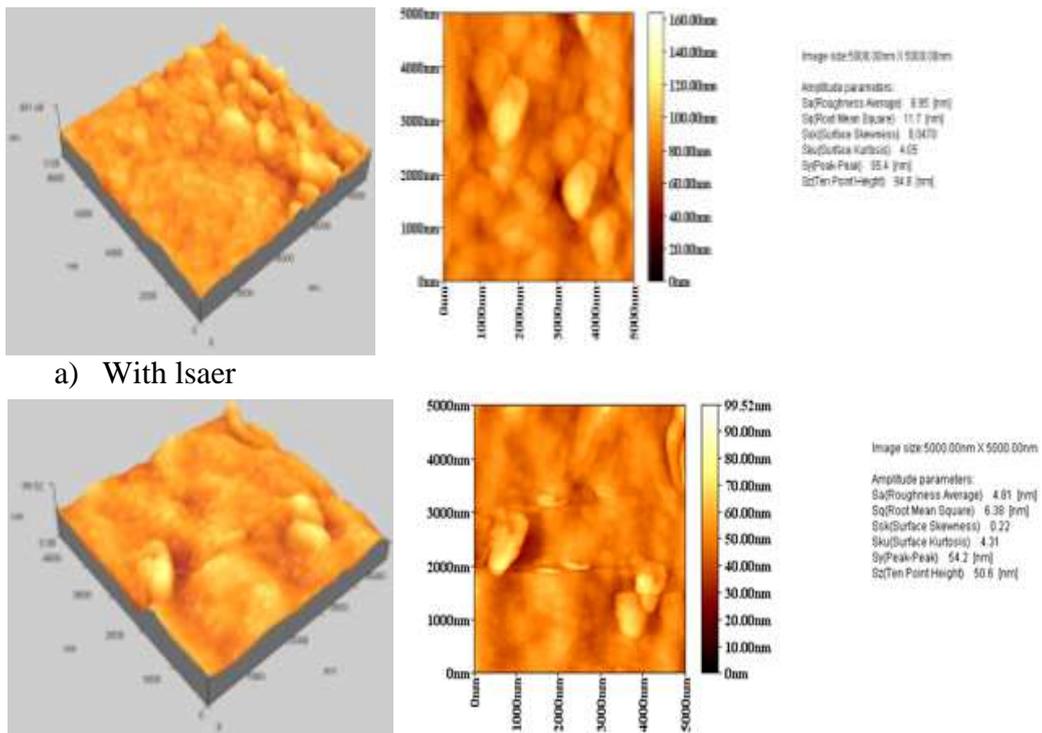


Figure (1) FTIR spectrum of PS/p-Si etching in different time.



a) With laser
 b) Without laser
 Figure (2) AFM image for stained samples a) with and b) without laser illumination for etching time 2 min.

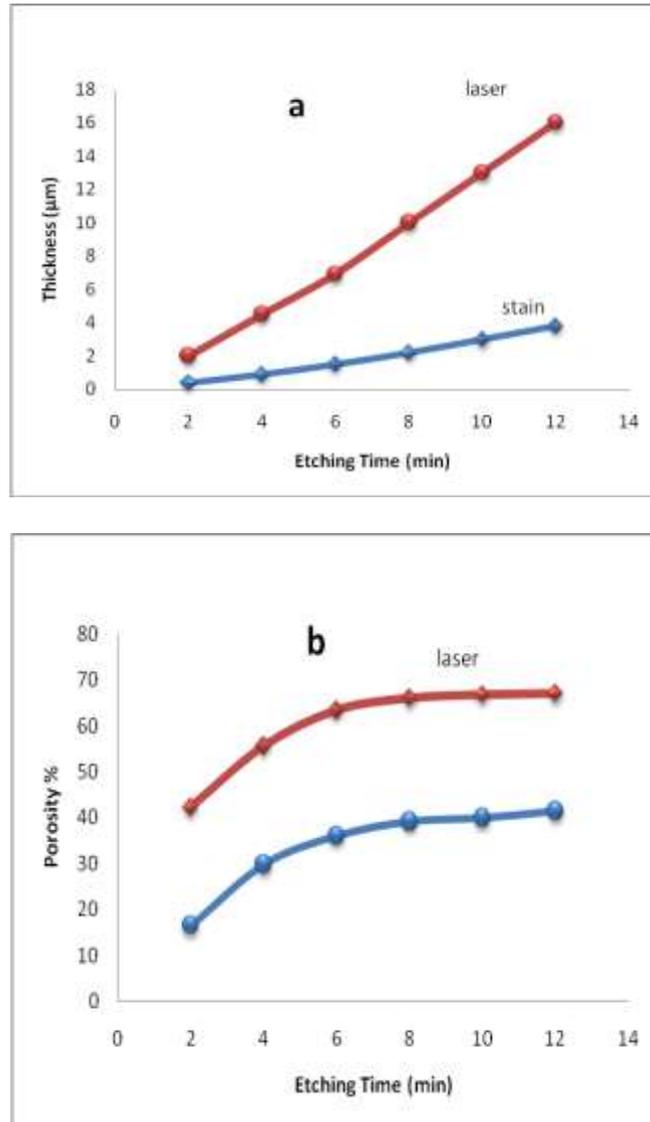


Figure (3) the relation between etching time with a) thickness and b) Porosity for prepared sample.

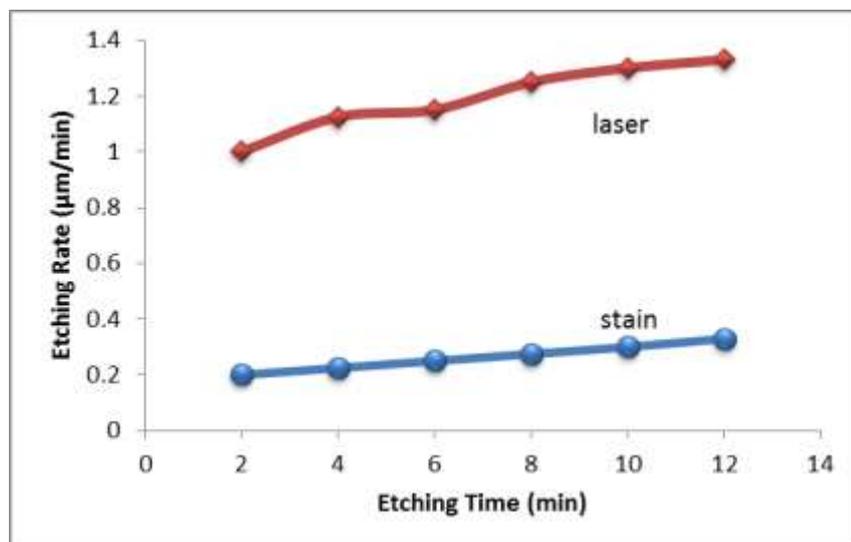


Figure (4) the variation of etching rate with etching time for prepared sample.